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An X-Ray Absorption Spectroscopy Study of Ball-Milled Lithium Tantalate and Lithium Titanate Nanocrystals

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Abstract. Previous work has shown that nanocrystalline samples of lithium tantalate and titanate prepared by high-energy milling show unusually high lithium ion conductivity. Here, we report an X-ray absorption spectroscopy (XAS) study at the Ti K-edge and the Ta L₃ edge of samples that have been milled for various lengths of time. For both systems the results show that milling creates amorphous material whose quantity increases with the milling time. The more extensive data for the tantalate shows that milling for only 30 minutes generates ~25% amorphous content in the sample. The content rises to ~60% after 16 hours. It is suggested that it is the motion of the lithium ions through the amorphous content that provides the mechanism for the high ionic conductivity.

1. Introduction

There are numerous reports in the literature that atomic migration in nanocrystals is unusually fast compared to the parent bulk material. In ionic materials the origin for enhanced diffusion has been assigned to atomic disorder along interfaces and surfaces [1-4], however, in many cases, the experimental data are ambiguous. The data for the binary and ternary oxides is particularly varied and is very dependent on the method of sample preparation [5-9] as this controls the microstructure of the material. A simple and rapid method of producing relatively large quantities of nanomaterials is high-energy ball-milling [10-13]. However, it is now clear that for simple binary oxides, like alumina, Al₂O₃ [14] and zirconia, ZrO₂ [15] it produces samples that have a large (tens of per cent) amorphous content; the nanocrystalline core is surrounded by a shell of amorphous material. The effect on the atomic migration of this microstructure is not clear. In contrast, ball-milled ternary oxides can show unusually enhanced ionic mobility.

The first ternary ball-milled system that was studied in any depth was lithium niobate, LiNbO₃, and the samples exhibited very high lithium ion diffusivity and conductivity [16-18];



the material in bulk crystalline form is normally regarded as an insulator. ^7Li nuclear magnetic resonance (NMR) studies of ball-milled LiNbO_3 with particle sizes down to 16 nm showed lithium ion mobility was increased by several orders of magnitude [16]. Nb K-edge X-ray absorption spectroscopy (XAS) of ball-milled LiNbO_3 showed a highly attenuated Extended X-ray Absorption Fine Structure (EXAFS) indicative of a significant amorphous content [17]. A very thorough study by Heitjans and co-workers [18] of ball-milled and sol-gel prepared LiNbO_3 using a wide range of techniques confirmed the amorphous nature of the surface of the nanocrystals and that this was the origin of the high lithium ion mobility.

Currently there is intense interest in improving the performance of lithium-ion batteries and a necessary requirement of components is high lithium ion mobility [19]. Hence the observations for ball-milled LiNbO_3 have led to the study of other similar materials. Ball-milled LiTaO_3 with a particle size of about 20 nm exhibits an ionic conductivity five orders of magnitude higher than bulk crystals [20]. Similarly ball-milling Li_2TiO_3 [21, 22] increases the lithium ion conductivity by some three orders of magnitude. As in the case of LiNbO_3 the increased lithium ion diffusion is assigned to the generation of amorphous material. In this contribution we confirm and quantify the presence of amorphous material in these two materials using XAS measurements.

2. Methodology

2.1 Materials

Micron sized commercial powders of Li_2TiO_3 and LiTaO_3 were ball-milled using the procedures described in previous papers [20-22]. For Li_2TiO_3 only two samples were studied; micron powder and after 8 hours milling. In the case of LiTaO_3 micron powder and samples after 0.5, 1, 4, 8 and 16 hours of milling were investigated. For the XAS measurements the powders were mixed with cellulose diluent and pressed into 13 mm diameter pellets.

2.2 XAS measurements

XAS scans were collected for the appropriate edge (Ti K-edge and Ta L_3 -edge) at room temperature on beam line B18 at the Diamond Light Source [23]. Data collection used transmission mode with ion chamber detectors. Continuous scanning (QEXAFS) was employed; an individual scan required 180 s and several scans were performed to improve the signal-to-noise ratio. The synchrotron energy and current were 3 GeV and 300 mA, respectively. The beam size at the sample

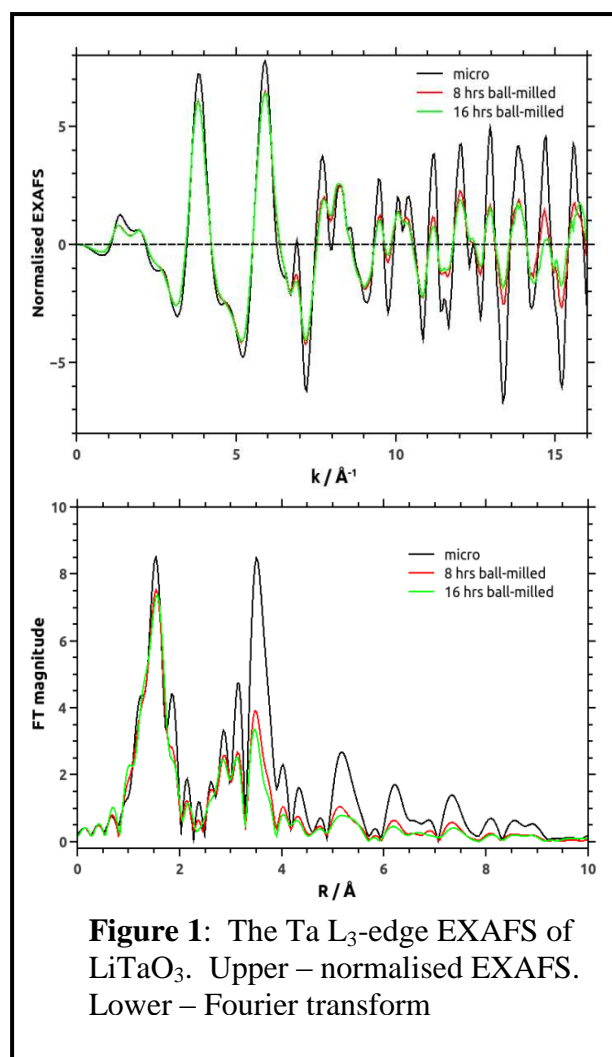


Figure 1: The Ta L_3 -edge EXAFS of LiTaO_3 . Upper – normalised EXAFS. Lower – Fourier transform

was 700×700 microns. Powdered samples were mixed with cellulose as a diluent and pressed into 13 mm diameter pellets. The spectra were normalized in Athena and fitted to scattering models in R-space produced by FEFF in Artemis [24].

3 Results and discussion

3.1 LiTaO₃ XAS measurements

All the spectra were analysed to yield the details of the local structure, i.e. bond lengths, R, Debye-Waller factors, σ , etc.; the qualitative information is, however, best represented by the Fourier transform (FT) of the k^3 weighted normalised EXAFS and these will be mostly used in this short paper. The data for three samples are shown in Figure 1. There is an attenuation of the EXAFS at high k leading to a reduction in the peaks in the corresponding Fourier transform, particularly the Ta-Ta correlation at 3.8 \AA . We have analysed the effect in two ways; as a change in the Debye-Waller factor due to disorder and as the effect of an amorphous content. A full analysis with FEFF allowing all the key parameters to float gave very good fits to the spectra. The parameters that changed significantly with milling time were the Debye-Waller factors, σ , for the Ta-O, Ta-O and Ta-Ta correlations at $1.9, 2.0$ and 3.8 \AA , respectively. These are plotted in Figure 2 in the form of σ^2 versus milling time. It can be seen that for all the correlations σ^2 almost doubles after only 30 minutes milling. This suggests that even this relatively brief milling causes a distortion of the local structure around the Ta ions. After 30 minutes the Ta-O Debye-Waller factors show only a gradual change whereas the Ta-Ta correlation, which is significantly larger, continues to increase steeply with milling time. An explanation of these effects is that milling after 30 minutes causes little changes to the local Ta-O octahedra but the relative arrangement of the octahedra continues to change. This would be expected given that the strong Coulomb interaction between Ta^{5+} and O^{2-} will maintain a relatively rigid local structure. This explanation is also consistent with the shape of the ^7Li NMR quadrupolar powder patterns observed for a sample milled for 30 min, see ref. [20]. After 30 min of milling most of the local electric field gradients in the direct neighbourhood of the Li nuclei are still comparable to those in unmilled, coarse-grained LiTaO₃ [20].

An alternative, but not contradictory, approach is to assume the attenuation of the EXAFS is due to the ball-milling generating amorphous material, as in the analysis of the EXAFS of ZrO₂ [15]. This would be mainly reflected in the peak height of the Ta-Ta shell in the Fourier transform. Assuming that the commercial powder prior to milling is 100% crystalline

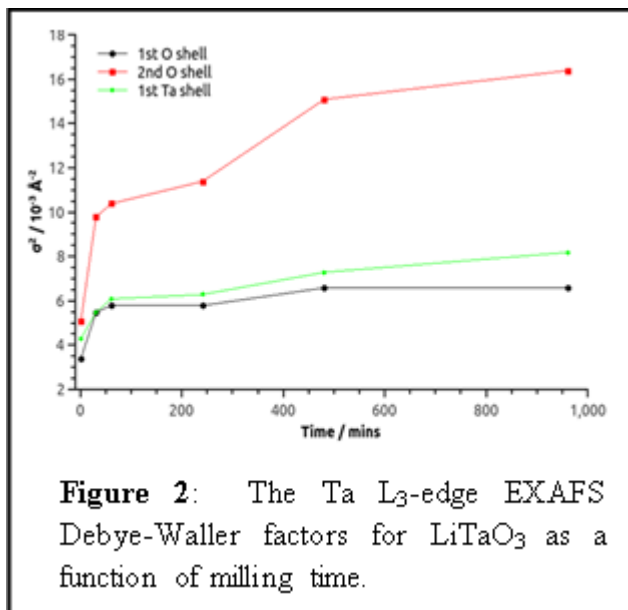


Figure 2: The Ta L₃-edge EXAFS Debye-Waller factors for LiTaO₃ as a function of milling time.

Milling Time/Hour	Estimated Amorphous Content	
	LiTaO ₃	Li ₂ TiO ₃
0	0%	0%
0.5	~ 25%	
1	~ 41%	
4	~ 44%	
8	~ 55%	~25%
16	~ 60%	

then the height of the Ta-Ta peak in the Fourier transform is representative of a sample with no amorphous content. Hence the relative height of this peak in the milled samples can be used to estimate the degree of crystallinity, i.e. the amorphous content. The results of this rather crude, but useful, analysis are shown in Table 1. This analysis also reveals that milling for as little as 30 minutes drastically affects the sample and generated ~25 % amorphous material.

3.2 Li_2TiO_3 XAS measurements

We have fewer data for this system but they follow the same pattern as those for LiTaO_3 . The EXAFS results are shown in Figure 3. Again, ball-milling attenuates the EXAFS and there is a decrease in the height of the Ti-Ti peak in the Fourier transform. The same analysis used for LiTaO_3 suggests the ball-milled sample contains 25% amorphous material.

It would be unwise to assume that the apparent quantitative differences between the titanate and tantalate after 8 hours ball-milling are meaningful. Other factors, such as the relative hardness of the tow materials, would need to be considered.

4. Conclusions

There are clear similarities between the current XAS results for LiTaO_3 and Li_2TiO_3 and the data for LiNbO_3 reported in earlier papers [17, 18]. Hence, a reasonable general explanation of the results for these two systems is that ball-milling is creating nanoparticles that have a crystalline core with an amorphous shell.

The more extensive study of LiTaO_3 is particularly interesting as it shows that ball-milling for only 30 minutes dramatically attenuates the EXAFS spectrum, indicating a highly disordered sample. This parallels the conductivity measurements of a sample ball-milled for 30 minutes where the magnitude is four orders higher than that for an un-milled sample [20]. However, the conductivity studies revealed some subtle dependence of the microstructure on the time of ball-milling. The longer the milling time the more difficult it is to recover crystallinity by thermal annealing.

Finally, it is interesting to compare the results for the current systems and other oxides with similar data for binary fluorides. XAS studies of ball-milled binary fluorides, like CaF_2

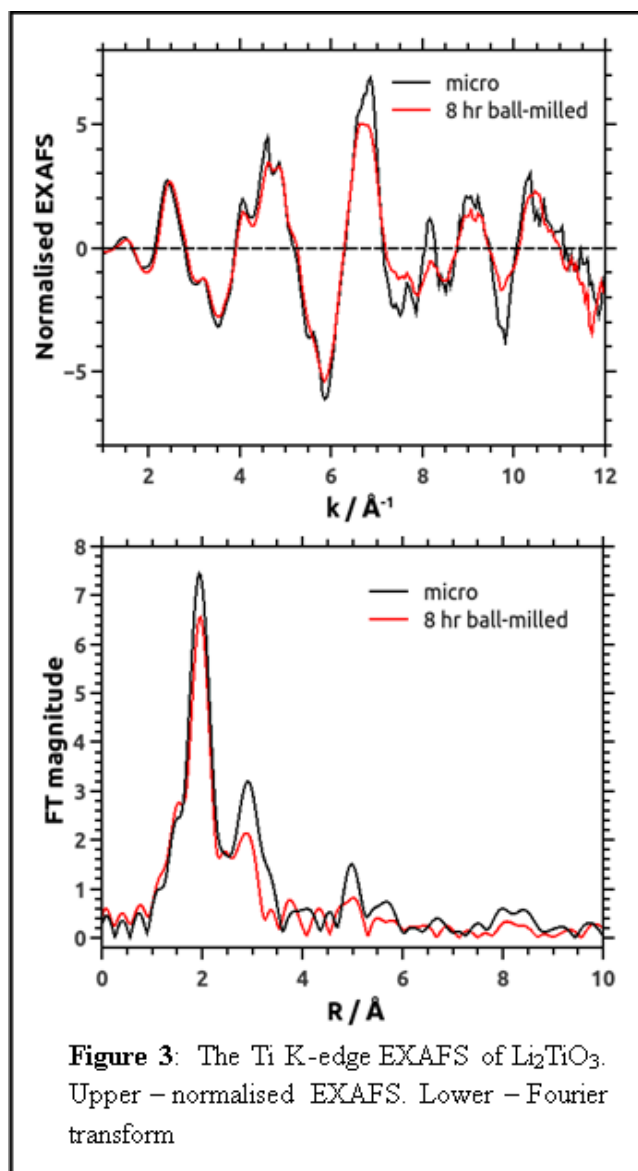


Figure 3: The Ti K-edge EXAFS of Li_2TiO_3 . Upper – normalised EXAFS. Lower – Fourier transform

and BaF₂ show virtually no attenuation of the EXAFS spectra, suggesting a very small, if any, amorphous content [25]. The difference is presumably a reflection of the greater strength of metal-fluorine bonds compared to metal-oxygen bonds.

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